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Structural and optical properties of nanocrystalline α -MoO₃ thin films prepared at different annealing temperatures

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Abstract Nanocrystalline α -MoO₃ thin films were prepared successfully by thermal annealing of molybdenum (Mo) thin films deposited on quartz and silicon substrates using DC magnetron sputtering method. The influence of annealing temperatures ranging from 400 to 1,000 °C on the structural, morphological and optical properties of the prepared films was investigated by X-ray diffraction, Fourier transform infrared spectrophotometer (FTIR) atomic force microscopic and UV–vis spectroscopy, respectively. The results show that the crystallinity and surface morphology of the films are strongly dependent on the annealing temperature. Also, the optimum annealing temperature of Mo films in our experiment was 600 °C and the films formed at this temperature exhibit only the (0*k*0) reflections and indicated the layered structure of α -MoO₃. The FTIR spectra confirm the formation of MoO₃. The transmittance of the MoO₃ films on quartz substrate was improved with increasing annealing temperature.

Keywords MoO₃ · Thin film · Layered structure · Annealing temperature · Nanocrystalline

Introduction

Molybdenum oxide (MoO₃) is a wide band gap *n*-type semiconductor material and one of the transition metal oxides. Some of the molybdenum oxide compounds are

thermodynamically stable normal phase of orthorhombic (α -MoO₃), metastable phase of monoclinic (β -MoO₃) and the hexagonal phase (h-MoO₃) [1–8]. The orthorhombic MoO₃ has a layered structure, containing two layers of octahedral MoO₆, held together by double covalent forces in the *a* and *c* axes; contiguous layers are kept together along the *b* axes by Van der Waal's forces [6]. MoO₃ thin films have attracted much interest for technological and industrial applications in recent years, including elements for energy efficient window technology, optical switching, coating high-density, memory devices, gas sensors, and lithium ion batteries [1–12, 15]. In addition, the optical property of MoO₃ is higher than those of many other inorganic materials because the incorporator indicates stronger and more uniform absorption of light in its colored state [13]. Different deposition techniques have been used for the growth of molybdenum oxide films on various substrates. Nirupama and Uthanna [2] have fabricated MoO₃ thin films on glass and silicon substrates using DC magnetron sputtering technique and investigate the influence of sputtering powers on the structural, electrical and optical properties of prepared films. Pulsed laser deposition method was employed by Al-Kuaili et al. [3] for preparation of MoO₃ thin films. Pardo and Torres [4] have synthesized MoO₃ thin films on silicon and glass substrates at different substrate temperature and annealing temperature. Yang et al. [5] have fabricated nanocrystalline MoO₃ thin films at various substrate temperatures by electron beam evaporation. Navias et al. [7] have fabricated MoO₃ nanorods by RF magnetron sputtering of molybdenum oxide target and study the structural properties. Flow rate effect on the structure and morphology of molybdenum oxide nanoparticles deposited by atmospheric pressure microplasma processing was studied by Bose et al. [8]. However, there are still only a few reports on the study of the preparation of MoO₃

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nanostructures on thin films. In the present investigation, the thermal oxidation of molybdenum (Mo) thin films was employed for the preparation of MoO₃ thin films. The effect of annealing temperatures on structural, surface morphological and optical properties of the films on quartz substrates are elaborately studied and presented here.

Experimental procedure

Firstly, Mo thin films were deposited on quartz and silicon substrates (1 cm × 1 cm) by dc magnetron sputtering (EDS-160) technique at room temperature. The purity of molybdenum target was 99/999 %. The chemical pretreatment of substrates was carried out with acetone and ethanol successively in ultrasonic bath for 10 min and dried in N₂ gas before introducing into the deposition chamber. The sputter chamber was evacuated to an ultimate pressure of 4×10^{-5} mbar using diffusion and rotary pumps before sputtering. The pure argon gas was admitted into the chamber and the working pressure was maintained at $\sim 2 \times 10^{-2}$ mbar throughout the depositions process. Before deposition, the target was pre-sputtered for 15 min to remove any surface contamination layer. The distance between the target and the substrate was kept at 7 cm. The cathode voltage and discharge current were 300 V and 180 mA, respectively. The thickness of the films and the rate of deposition were controlled using the quartz crystal monitor in the chamber. The rate of sputtering (0.1 nm s^{-1}) was used to deposit all Mo films to a thickness of 140 nm. During the deposition, the substrates are not intentionally heated. In the second step, thin films were annealed in the electrical furnace at various annealing temperature of 400, 600, 800, 1,000 °C under constant oxygen flow for 1 h. The flow rate of the O₂ gas (purity 99/999 %) was 80 sccm. Temperature variations are imparted very slowly to avoid additional stresses or cracks in the films. The as-deposited films and those formed during thermal oxidation at temperatures of 400, 600, 800 and 1,000 °C are abbreviated as Q0, Q1, Q2, Q3 and Q4, respectively. The films formed on silicon substrates were analyzed with Fourier transform infrared spectrophotometer (FTIR, Perkin Elmer spectrum 100) in the wave number range of 400–1,200 cm⁻¹ to determine the chemical binding configuration. The crystal structure of the films was characterized by X-ray diffraction (XRD, Philips, pw 1800) using CuK α radiation ($\lambda = 0.1506 \text{ nm}$) at 40 kV and 30 mA in the 2θ scan ranging from 10 to 90. The morphology of MoO₃ thin films was studied by atomic force microscopic (AFM, park Scientific Instrument, Auto probe cp USA). The transmittance spectra of the MoO₃ thin films deposited on the quartz substrates were recorded using a spectrophotometer.

Results and discussion

The XRD patterns of the Mo thin films and MoO₃ films prepared at different annealing temperatures are shown in Figs. 1, 2, 3, 4 and 5. The as-deposited film (sample Q0) shows that three peaks belong to Mo crystalline structure in (110), (211), (220) crystallography directions. Obviously, the Mo (110) peak is dominant. By annealing Mo films at temperature of 400 °C (sample Q1), the polycrystalline MoO₃ can be observed. By increasing the annealing temperature to 600 °C (sample Q2), the film showed only (0k0) reflections related to orthorhombic α -phase of molybdenum oxide. The presence of (0k0) reflections indicated the layered structure of α -MoO₃. In addition, the high intense and sharp peaks observed in the XRD pattern of this films confirmed the highly oriented and

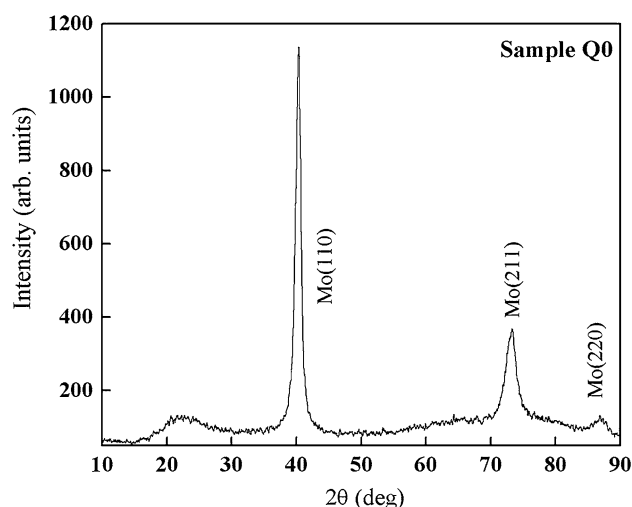


Fig. 1 The XRD pattern of the Mo film deposited on quartz substrate

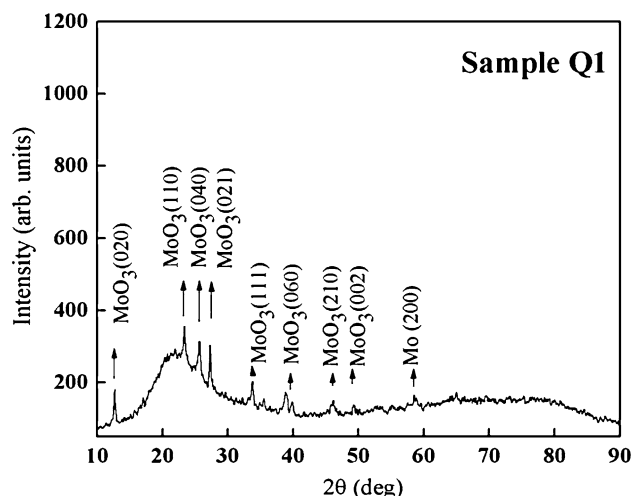


Fig. 2 The XRD pattern of the MoO₃ film prepared at annealing temperature of 400 °C



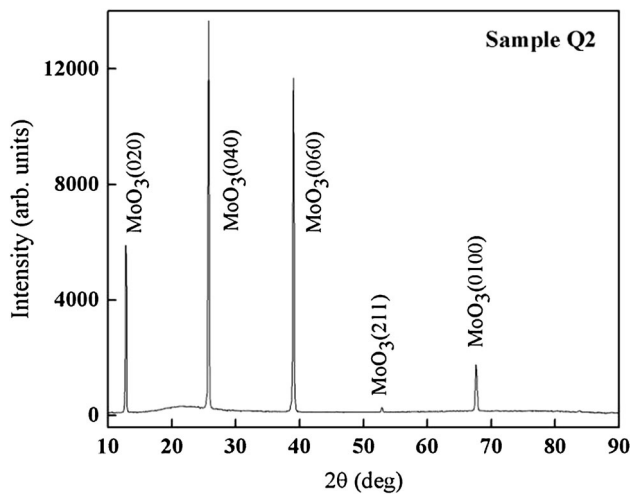


Fig. 3 The XRD pattern of the MoO₃ film prepared at annealing temperature of 600 °C

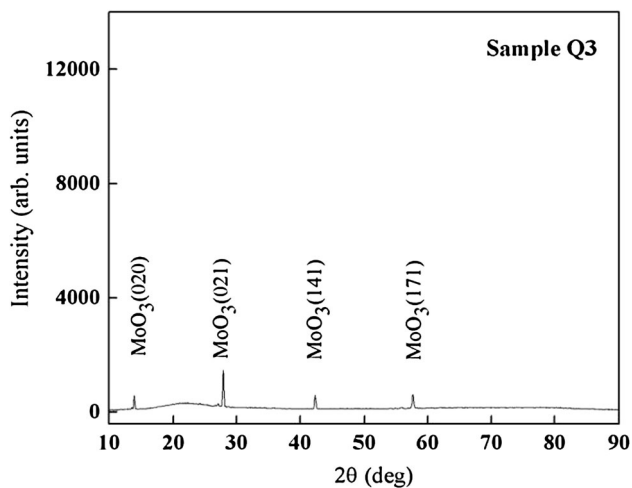


Fig. 4 The XRD pattern of the MoO₃ film prepared at annealing temperature of 800 °C

polycrystalline nature of the film. In XRD pattern of sample Q3 that was prepared at temperature of 800 °C, the crystalline properties were reduced in comparison with sample Q2, because the number and intensity of peaks were decreased. Finally, for sample Q4 the crystallinity of MoO₃ film was deteriorated maybe for disassembled the lattice arrangement and the structure of film is amorphous. The XRD results show that the optimum temperature of thermal oxidation of Mo films for preparation of single phase of orthorhombic α -phase MoO₃ film in our experiment was 600 °C, because in this temperature the best crystallinity was observed.

The average crystalline size and microstrain of the prepared films have been obtained from the following Scherrer relations [14]:

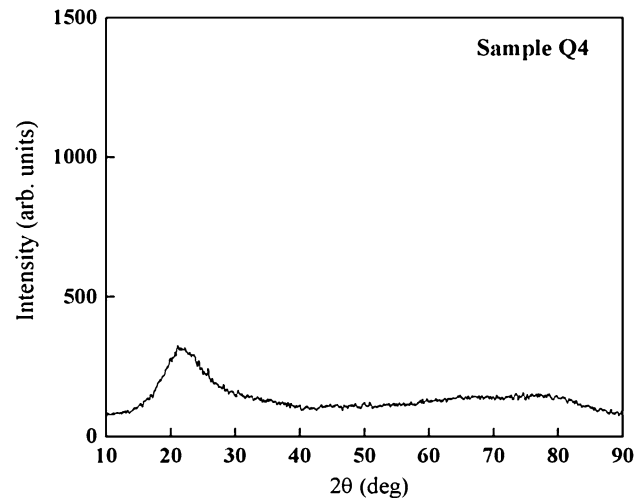


Fig. 5 The XRD pattern of the MoO₃ film prepared at annealing temperature of 1,000 °C

$$D_{hkl} = 0.9\lambda / \beta_{hkl} \cdot \cos \theta \quad \text{and} \quad \varepsilon = \beta / 4 \tan \theta. \quad (1)$$

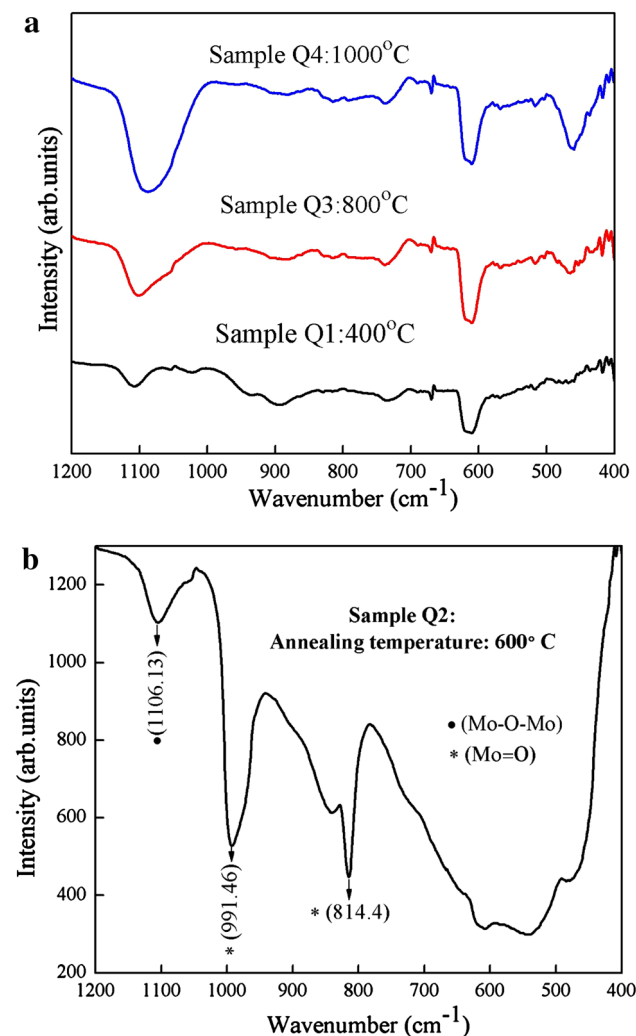
Here, λ is the X-ray wavelength (1.5406 nm for CuK α), θ is the Bragg diffraction angle and β_{hkl} is the full-width at half-maximum (FWHM) in radians of the main peak in the XRD pattern. The average crystal sizes for all samples are shown in Table 1. The average crystalline size of the MoO₃ films calculated from the Scherrer's formula was in the range of 24.34–41.74 nm, which confirms the presence of nanocrystals in the films prepared in this study. Also, we can observe that annealing temperature augmentation up to 600 °C leads to increase in average crystal size and decrease in microstrain. The increase in crystallite size is related to decrease in strain.

The Fourier transform infrared spectra of MoO₃ thin films formed on silicon substrate at different annealing temperatures are shown in Fig. 6a, b. The films prepared at annealing temperatures of 400, 800 and 1,000 °C have similar spectra and three peaks can be observed at 612.26, 737.6 and 1,108.26 cm⁻¹. The peak of 612.26 cm⁻¹ corresponds to silicon substrate and two other peaks are belonging to molybdenum oxide films. By increasing the annealing temperature, the intensity of the peaks is increased. The films annealed at 600 °C have a different FTIR spectrum and exhibit characteristic peaks at 814.4, 993.6 and 1,108.26 cm⁻¹ that two intense peaks seen at 814.4, 993.6 cm⁻¹ were attributed to the Mo=O stretching vibration, which is an indicator for the layered orthorhombic MoO₃ phase. These results confirm the results obtained by XRD analysis. The FTIR studies of the present work agree well with the reports on MoO₃ thin films prepared using dc magnetron sputtering method by Uthanna et al. [15].



Table 1 Comparison of MoO₃ films' structural parameters

Thermal annealing temperature (°C)	Sample name	2 θ (°)	Phase	(hkl)	FWHM $\times 10^{-3}$	Average crystalline size D (nm)	Micro strain (ϵ) $\times 10^{-2}$
–	Q0	40.38	Mo	(110)	16.7	9.21	1.14
400	Q1	12.75	MoO ₃	(020)	4.71	30.95	1.05
		23.40		(110)	4.71	31.38	0.56
		25.68		(040)	6.10	24.34	0.66
		27.32		(021)	4.71	31.63	0.48
		27.32		(021)	4.71	31.63	0.48
600	Q2	12.84	MoO ₃	(020)	3.49	41.74	0.77
		25.77		(040)	3.66	40.57	0.40
		39.06		(060)	3.84	39.99	0.27
800	Q3	13.87	MoO ₃	(020)	3.54	41.19	0.72
		27.91		(021)	3.91	38.15	0.39
		42.42		(141)	4.41	35.17	0.28
		57.64		(171)	5.58	29.61	0.25

**Fig. 6** **a** The FTIR spectra of the MoO₃ film prepared at different annealing temperatures. **b** The FTIR spectrum of the MoO₃ film prepared at annealing temperature of 600 °C

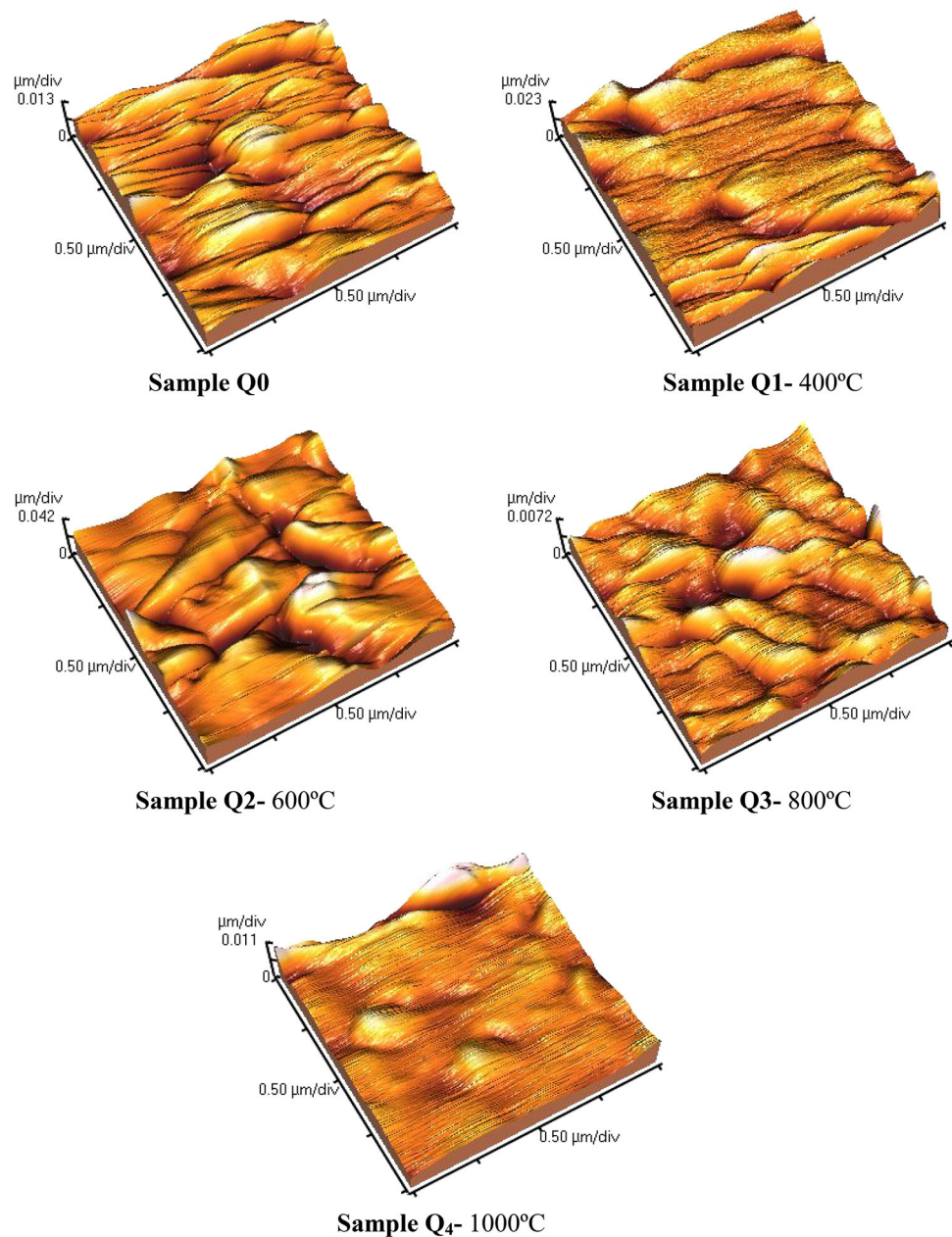
The three-dimensional AFM images of the prepared films are shown in Fig. 7. The AFM images show the features of structural change and grain growth for MoO₃ thin films on quartz substrates at various annealing temperatures.

The AFM images show the growth of grain and smooth surface for sample Q0 and by thermal oxidation, the grains combine and make bigger grains and roughness of film increases. The increase in films' surface roughness probably attributed to the coalescence of grains in the film structure that leads to the formation of large grain aggregates. Also, the layered structure of the MoO₃ films is clearly seen in AFM images and the AFM results are in good agreement with the results obtained by the X-ray analysis. The plot of surface roughness versus thermal annealing temperatures is shown in Fig. 8. It can be considered that the surface roughness of the films is dependent mainly on surface morphology. Changes in rms roughness of the films may also due to the different crystal structure of MoO₃ thin films at various annealing temperatures [16].

Optical properties

The as-deposited film (sample Q0) is dark, whereas those annealed at different oxidation temperatures are almost different colors, such as deep blue (sample Q1), light blue (sample Q2), grayish/blue (sample Q3) and transparent (sample Q4). The influence of thermal oxidation temperature on the optical transmittance of MoO₃ films deposited on quartz substrate was studied in the wavelength range between 200 and 800 nm and the recorded spectra are shown in Fig. 9a, b. Figure 9a shows that the MoO₃ film prepared at temperature of 400 °C has no transmittance

Fig. 7 3D AFM images of the MoO_3 thin films at different annealing temperatures



(0 %) and for thermal oxidation temperature of 600 °C, very low transmittance (1.5 %) can be observed. This can be due to the large scattering introduced by the enhanced roughness in these films, as revealed by AFM (Fig. 7). By enhancement of thermal annealing temperature, the MoO_3 film transmittances in the wavelength range of 200–800 nm change from 30 to 70 % for temperature of 800 °C and 35 to 80 % for temperature of 1,000 °C, respectively (Fig. 9b). These results show that the transmittance of MoO_3 films was very sensitive to the annealing temperatures and increasing annealing temperature leads to an improvement in the optical transmittance [17].

Conclusion

Thin films of MoO_3 were prepared by thermal annealing of Mo films deposited on quartz and silicon substrate using dc magnetron sputtering method. The annealing temperatures were varied between 400 and 1,000 °C. The XRD studies revealed that the films structure changes from crystalline to amorphous by enhancing the annealing temperature. Also, the best crystallinity and layered structure were observed at annealing temperature of 600 °C. The layered structure of the MoO_3 films is clearly seen in AFM images. Also, the AFM and FTIR results are in good agreement with the



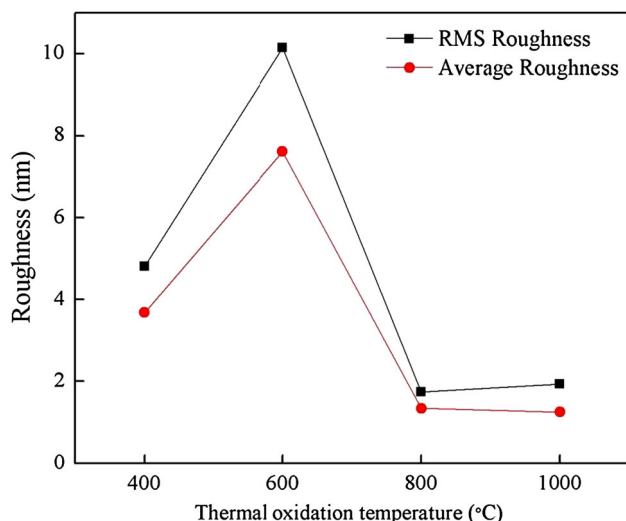


Fig. 8 The films roughness versus thermal oxidation temperatures

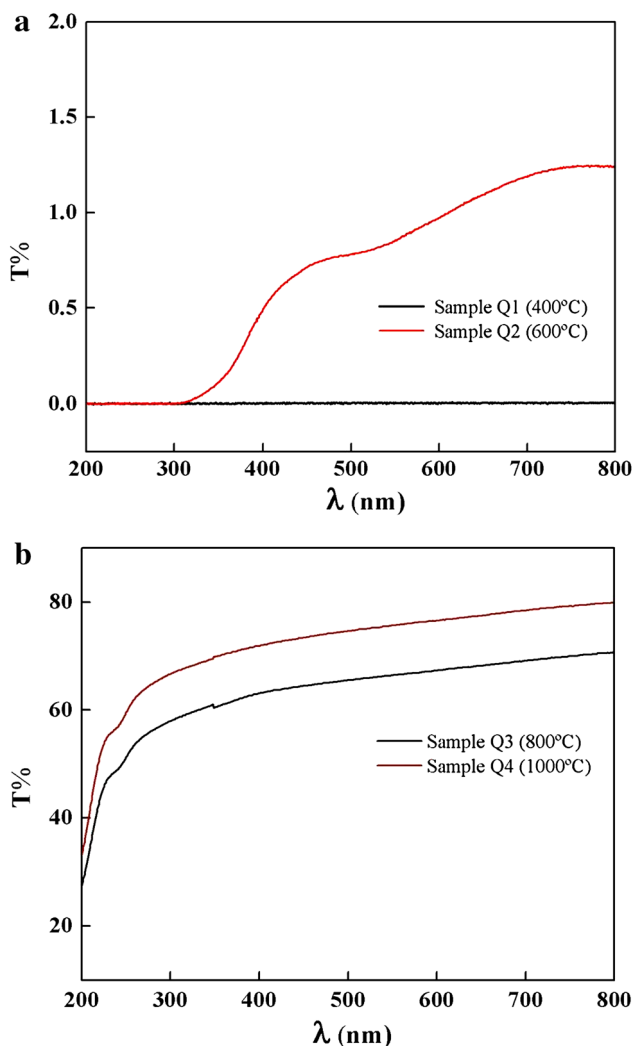


Fig. 9 Transmittance spectra of the MoO₃ thin films prepared at different annealing temperatures. **a** 400, 600 °C and **b** 800, 1,000 °C

results obtained by the X-ray analysis. The optical transmittance of the MoO₃ thin films as a wavelength indicated that the annealing temperature strongly affected the optical transmittance of films and the good transparency was observed at annealing temperature of 800 and 1,000 °C.

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